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Advanced Powder Technology xxx (2016) xxx-xxx

Contents lists available at ScienceDirect Advanced Powder Technology

journal homepage: www.elsevier.com/locate/apt

Original Research Paper

Dynamic modeling and Molecular Weight Distribution of ethylene copolymerization in an industrial gas-phase Fluidized-Bed Reactor

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ARTICLE INFO

Article history:
 Received 21 December 2015
 Received in revised form 19 April 2016
 Accepted 18 May 2016
 Available online xxxx

22 Keywords:23 Polyethylen

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19

23 Polyethylene 24 Molecular Wei

24 Molecular Weight Distribution 25 Eluidized Bed Reactor

5 Fluidized-Bed Reactor

26 Dynamic modeling27 Solid elutriation

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ABSTRACT

A dynamic model for ethylene copolymerization in an industrial Fluidized-Bed Reactor (FBR) is developed to describe its behavior and calculate the properties of polyethylene. The presented model considers particle entrainment and polymerization reaction in two phases. Two-site kinetic and hydrodynamic models in combination, provide a comprehensive model for the gas phase fluidized-bed polyethylene production reactor. The governing moment and hydrodynamic differential equations were solved simultaneously and the results compared with a similar work, as well as industrial data. The dynamic model showed accurate results for predicting Polydispersity Index (PDI), Molecular Weight Distribution (MWD), reactor temperature and polymer production rate.

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42 1. Introduction

Olefin polymerization in gas-phase Fluidized-Bed Reactors
(FBR) has been recognized as one of the most economic methods
of manufacturing commodity polymers including polyethylene
(PE), polypropylene (PP) and ethylene-propylene rubber (EPR).
Union carbide, commercialized the first gas-phase fluidized-bed
polymerization process, i.e., UNIPOLTM* Process, to produce polyethylene in 1986 [1].

Chemical processes such as gas-solid reactions or gas-phase 50 51 reactions catalyzed by solids are among the operations which FBRs 52 are utilized extensively. A common use of FBRs in industry is to 53 produce linear low-density polyethylene (LLDPE) by employing 54 heterogeneous Ziegler–Natta (Z-N) catalysts. In contrast to other 55 processes used to produce polyethylene, polymerizing ethylene 56 in a gas-phase FBR have advantages such as better heat removal, operating at decreased temperatures and pressures, and not 57 requiring solvents, which help make it the most broadly used pro-58 cess in industries [2]. Fig. 1 shows a typical fluidized-bed polyethy-59 60 lene reactor process flow diagram.

Fresh feed mixture consisting of ethylene as monomer, 1butene as co-monomer, hydrogen, and nitrogen is injected from

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the bottom of the reactor and enters the reaction zone via a distributor. The *Z*–*N* catalyst particles are introduced continuously above the distributor to activate the reactants. After fluidization, unreacted gases are separated in the disengagement part of the reactor. Entrained solids carried by the gas are separated in the cyclone and gets recycled back into the reactor. The reacted gas then passes through the compressor and heat exchanger to be mixed with fresh feed and gets recycled back into the reactor. The product leaves the reactor from just above the distributor and gets collected in a cylinder. Normally, polyethylene producing FBRs available in the industry work within a temperature range of 75–110 °C and pressure range of 20–40 bar [3].

The amount of superficial gas velocity (U_0) can be somewhere between 3 and 8 times the minimum fluidization velocity. Various models have been suggested to predict how a gas-phase ethylene polymerization perform in the real world. Researchers have modeled these FBRs as single, two or three phase reactors [4–6]. As a result of assuming that bubbles are free from solids, all of these models considered that reactions occur only in the emulsion phase. Jafari et al. [7] compared the performance of some available models at the time, such as the simple two phase model, dynamic twophase model and generalized bubbling/turbulent model. They concluded that the bubbling/turbulent model results are better fitted to experimental data in comparison with other approaches.

An overview on modeling different scales available in multiphase chemical reactors such as the heat and mass transfer,

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Nomenc	lature	
A	cross sectional area of the reactor (m^2)	<i>/</i> /\\
$[M_i]$	concentration of component <i>i</i> in the reactor (kmol/m ³) n	าพ
$[M_i]_{in}$	concentration of component <i>i</i> in the inlet gaseous <i>N</i>	<i>I</i> (0
	stream N	I*(j
AlEt ₃	triethyl aluminum co-catalyst N	I _d (j
Ar	Archimedes number N moles of reacted monomer bound in the polymor in the	l _{dIF} 1
Di	reactor	'H
CFD	computational fluid dynamics N	I _i (r
$C_{p,pol}$	specific heat capacity of solid product (J/kg K)	J.
C_{pg}	specific heat capacity of gaseous stream (J/kg K) P	•
C_{pi}	specific heat capacity of component <i>i</i> (J/kg K) P	DI
C_{pMi}	specific neat capacity of the component i (J/Kg K) Q	<u>l(</u> r,
d_{b0}	initiate bubble diameter (m)	
D_{g}	gas diffusion coefficient (m^2/s)	e _m
d_p	particle diameter (m)	
D_t	reactor diameter (m) R	i
FBR	Fluidized-Bed Reactor R	р
r _{cat} fi	fraction of total monomer in the reactant gas which is T	v
Ji	monomer M_i t	
g	gravitational acceleration (m/s ²)	in
GPC	Gel Permeation Chromatography T	ref
Н	height of the reactor (m)	J_t^*
H ₂	hydrogen U	0 1
п _{bc} Н.	bubble to cloud heat transfer coefficient ($W/m^3 K$) U	'b I.
H _{ce}	cloud to emulsion heat transfer coefficient (W/m^3 K)	br J _{mf}
HDPE	high-density polyethylene	J_t
i	monomer type	r
I _m	impurity such as carbon monoxide (kmol/m ³)	p
J K.	active site type v elutriation constant in hubble phase (kg m ² s ⁻¹)	V _b V
K_{bc}	bubble to cloud mass transfer coefficient (s^{-1}) X	(n
Kbe	bubble to emulsion mass transfer coefficient (s ⁻¹)	
K _{ce}	cloud to emulsion mass transfer coefficient (s^{-1}) Y	'(n,
$k_{dI}(j)$	deactivation by impurities rate constant for a site of	
k(i)	type J Z spontaneous deactivation rate constant for a site of type	- <i>r</i>
(ast)	<i>i</i>	re
K _e	elutriation constant in emulsion phase (kg m ² s ⁻¹)	\H
$k_f(j)$	formation rate constant for a site of type j δ	
$k_{fhi}(j)$	transfer rate constant for a site of type j with terminal ϵ_j	b
k_{i} (i)	monomer M_i reacting with nydrogen \mathcal{E}_i	е
Kfmi()	monomer M_i reacting with monomer M_k	mf I
$k_{fri}(j)$	transfer rate constant for a site of type j with terminal).)
	monomer M_i reacting with AlEt ₃ ρ	g vol
$k_{fsi}(j)$	spontaneous transfer rate constant for a site of type j ϕ	s
l.	with terminal monomer M_i	
$k_g(i)$	rate constant for reinitiating of a site of type <i>i</i> by mono-	ub.
$\kappa_{m}(\mathbf{y})$	mer M_i	
$k_{hr}(j)$	rate constant for reinitiating of a site of type <i>j</i> by cocat-	
	alyst e	
ki _i (j)	rate constant for initiation of a site of type j by mono- i	
$kn_{i}(i)$	$\frac{j}{j}$	
∿PikU)	nal monomer M_i reacting with monomer M_i	ıf
kp _{Ti}	propagation rate constant (m ³ /kmol s)	0l of
LLDPE	linear low-density polyethylene	⊑j `_T
MFI	melt flow index (g/10 min)	, .
Mn Mw	number average molecular weight of polymer (kg/kmol)	
1 V1 VV	weight average molecular weight of polyhiet (Kg/KIIIOI)	

MWD	Molecular Weight Distribution
mw;	molecular weight of monomer i (g/mol)
N(0, i)	uninitiated site of type <i>j</i> produced by formation reaction
$N^*(j)$	potential active site of type <i>j</i>
$N_d(j)$	spontaneously deactivated site of type <i>j</i>
$N_{dIH}(0, j)$	impurity killed sites of type <i>j</i>
N _H	uninitiated site of type <i>j</i> produced by transfer to hydro-
	gen reaction
$N_j(r, j)$	living polymer molecule of length <i>r</i> , growing at an ac-
	tive site of type <i>j</i> , with terminal monomer <i>M</i>
P	pressure (Pa)
	Polydispersity index
Q(r, j)	twoe i
r	number of units in polymer chain
Re c	Reynolds number of particles at minimum fluidization
nemj	condition
Ri	instantaneous consumption rate of monomer (kmol/s)
R _n	production rate (kg/s)
R_{v}^{P}	volumetric polymer outflow from the reactor (m^3/s)
T	temperature (K)
t	time (s)
T _{in}	temperature of the inlet gaseous stream (K)
Tref	reference temperature (K)
U_t^*	dimensionless terminal falling velocity coefficient
U_0	superficial gas velocity (m/s)
	bubble velocity (m/s)
U _{br}	minimum fluidization velocity (m/s)
U _{mf} II.	terminal velocity of falling particles (m/s)
V	reactor volume (m ³)
V_n	volume of polymer phase in the reactor (m^3)
W_{b}	weight of solids in the bubble phase (kg)
We	weight of solids in the emulsion phase (kg)
X(n, j)	nth moment of chain length distribution for dead poly-
	mer produced at a site of type <i>j</i>
Y(n, j)	<i>n</i> th moment of chain length distribution for living poly-
7 N	mer produced at a site of type j
Z–N	Ziegler–Natta catalyst
Croal. 1.	
	ers
$\Delta \Pi_R$	volume fraction of hubbles in the bed
EL.	void fraction of bubbles in the bed
с _о	void fraction of emulsion for Geldart B particles
Emf	void fraction of the bed at minimum fluidization
μ	gas viscosity (Pa s)
ρ_g	gas density (kg/m ³)
ρ_{pol}	polymer density (kg/m ³)
ϕ_s	sphericity for sphere particles
C 1 ·	
Subscripts	s and superscripts
1 ว	ennymene 1. butene
∠ h	I-Dutelle hubble phase
р Р	emulsion phase
i	component type number
i	active site type number
mf	minimum fluidization
pol	polymer
ref	reference condition

T, TT pseudo kinetic rate constants



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Fig. 1. Schematic diagram of an industrial fluidized-bed polyethylene reactor.

kinetics and hydrodynamics was given by Bi and Li [8]. The authors 89 90 also proposed a Computational Fluid Dynamics (CFD) two fluid model which works towards minimizing energy in single-phase 91 flow. Choi and Ray [4] separated the reactor into two regions of 92 emulsion phase and bubble phase after McAuley et al. [6] regarded 93 94 the fluidized bed polymerization reactor to be a well-mixed reactor 95 or continuously stirred-tank reactor (CSTR). Fernandes and Lona [9] considered gas in bubble and emulsion phases plus solid poly-96 97 mer particles, all as plug flow phases, to propose their three-phase 98 heterogeneous model. Instead of considering constant bubble size, 99 Hatzantonis et al. [10] studied the dynamic and steady state 100 behavior of reactor when the bubble size varies. Besides breaking 101 down the reactor into two sections of emulsion phase and bubble phase, they also divided the bubble phase into N well-mixed sec-102 tions in series and assumed that the emulsion phase is seamlessly 103 104 mixed. The size of each section in their model, was fixed equivalent to the diameter of the bubble at the relative bed height. Zheng et al. 105 [11] developed a steady-state and dynamic methodology to model 106 107 the propylene process using the Spheripol Technology. Their 108 kinetic model was based on both single and multisite catalyst 109 and their Molecular Weight Distribution (MWD) results were fitted 110 using the actual Gel Permeation Chromatography (GPC) data. In a similar work, Luo et al. [12] developed a methodology to model 111 the polypropylene process based on Hypol Technology. The 112 authors used Polymer Plus and Aspen Dynamics to predict process 113 114 behavior and physical properties of the steady-state and dynamic modes. 115

Meanwhile, some researchers focused on particle size distribution studies in fluidized beds rather than kinetics or property estimation [13–15]. Furthermore, fluidization regimes have also been studied in many works. Different methods to determine fluidization regimes in gas-solid FBRs, such as radioactive particle track-120 ing, electrical capacitance tomography, or magnetic resonance 121 122 imaging have been applied on these reactors to study different 123 hydrodynamic aspects [16-18]. Alizadeh et al. [2] introduced a pseudo-homogeneous tanks-in-series model to predict the behav-124 ior of industrial-scale gas-phase polyethylene production reactor. 125 Kiashemshaki et al. [19] got inspiration from this model and pro-126 posed a two-phase model to describe the fluidized bed ethylene 127 polymerization reactor. Their model was a dynamic model except 128 in terms of calculating temperature and comonomer concentra-129 tions. Shamiri et al. [20-22] studied different dynamic modeling 130 and control approaches for gas phase homopolymerization or 131 copolymerization of olefin in FBRs. 132

In the current study, a fully dynamic modeling approach is used to predict the kinetic and hydrodynamic behavior of industrial polyethylene production reactors and polyethylene properties. The advantage of this model is that it is a two-phase model which considers reaction to take place in both phases. Furthermore, particle elutriation has also been considered in order to take the losses of entrained catalyst and polymer particles from the fluidized bed into account.

A two-site copolymerization kinetic scheme for ethylene and 1-butene were used in this study in order to gain a more realistic picture of copolymerization over a heterogeneous Z-N catalyst in a FBR. Then, the results were compared with a semidynamic two-phase model from literature to show the benefits of the current model in comparison to existing models and demonstrate how considering elutriation in ethylene polymerization modeling leads to more realistic results. Lastly, the modeling results have been validated by comparison with industrial data.

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151 2. Modeling

152 2.1. Polymerization mechanisms

The modeling approach depends on whether we discuss 153 homopolymerization or copolymerization. In homopolymerization, 154 155 only one monomer is involved in the production of the polymer, 156 while in copolymerization reaction, there are two types of mono-157 mer forming the polymer. In the current study, the kinetic model 158 developed by De Carvalho et al. [23] and McAuley et al. [24] was 159 employed to produce a comprehensive mechanism which describes the kinetics of copolymerization of ethylene and 1-160 161 butene catalyzed by two sites of the Z-N catalyst. Table 1 lists the reactions, comprising formation, initiation, propagation, trans-162 fer and deactivation of the active sites. To solve the equations, 163 method of moments was used. These related moments equations 164 are listed in Table 2. The index *i* in the tables refers to the type 165 166 of monomer and index *j* refers to the type of the active site. Table 3 gives the rate constants of each reaction for both site types that 167 were used in this work and mentions their sources in the literature. 168 If we assume that monomers are primarily consumed over the 169

propagation reactions, we can obtain the equation for consumption rate of each component. Eq. (1) shows this mathematical statement after solving the moment equations [24]:

$$R_k = \sum_{j=1}^{m} \sum_{i=1}^{m} [M_k] Y(0,j) k_{p_{ik}} \quad k = 1, 2, \dots$$
(1)

176where *m* is the number of each type of monomer and *ns* is the num-177ber of each type of active site. Then, we can get the total polymer178production from Eq. (2):

$$R_p = \sum_{k=1}^m m w_k R_k$$

182 2.2. Hydrodynamics

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2.2.1. Available models

Series and parallel reactions in catalytic polymerization of ethy-184 lene with alpha-olefin copolymers makes this process rather com-185 186 plicated. The extensive multi-site kinetic model proposed by McAuley et al. [24] considers copolymerization of olefins over 187 heterogeneous *Z*–*N* catalysts. The main fundamental reactions in 188 189 polymerization which were considered in this study have been 190 given in Table 1. These reactions include the formation of active 191 centers, insertion of monomers into the growing polymer chains, 192 chain transfer reactions, and catalyst de-activation respectively.

193 The method of moments is the most frequently used method for 194 modeling polymerization. This is because, by applying this method, we can foretell polymer properties such as density, Polydispersity 195 Index (PDI), average molecular weight, and branching frequency as 196 197 well as the ability to calculate operating variables, i.e., reactor tem-198 perature, rate of polymer production and rate of components con-199 sumption (monomers and hydrogen). These moment equations 200 have been given in Table 2.

Kiashemshaki et al. [19] used a semi-dynamic model to predict polymer properties in the sense that they did not produce dynamic temperature and comonomer concentrations profiles in their model. In order to model the gas-phase LLDPE production FBR, they divided the bed into several Plug Flow Reactors (PFR) and CSTRs in series. To model such a reactor, Kiashemshaki et al. [19] made several assumptions, which are as follows:

Table 1

Reactions occurring in a copolymerization reaction [24].

Description	Reaction
Formation reaction	$N^*(j) \stackrel{kf(j)}{\rightarrow} N(0,j)$
Initiation reaction	$N(0, i) + M_i \stackrel{ki_{i(j)}}{\longrightarrow} N_i(1, i)$ $i = 1, 2,$
Propagation	$N_{i}(r, i) + M_{i} \stackrel{kp_{k}(r, j)}{\longrightarrow} N_{i}(r+1, i), i-k-1, 2$
Transfer to monomer	$N_i(r, j) + M_k \stackrel{\text{ff}m_{k}(j)}{\longrightarrow} N_k(r+1, j) \stackrel{\text{f}}{\longrightarrow} = k = 1, 2, \dots$
Transfer to hydrogen	$N_{i}(r,j) + H_{2} \xrightarrow{kfh_{(j)}} N_{H}(0,j) + Q(r,j) i = 1, 2, \dots$
	$N_H(0,j) + M_i \stackrel{kh_{i(j)}}{ ightarrow} N_i(1,j) i=1,2,\ldots$
	$N_H(0,j) + \text{AlEt}_3 \stackrel{kh_r(j)}{\rightarrow} N_1(1,j)$
Transfer to co-catalyst	$N_i(r,j) + \text{AlEt}_3 \xrightarrow{kfr_i(j)} N_1(1,j) + Q(r,j) i = 1, 2, \dots$
Spontaneous transfer	$N_i(r,j) \stackrel{kf_{S(j)}}{\rightarrow} N_H(0,j) + Q(r,j) i = 1, 2, \dots$
Deactivation reactions	$N_i(r,j) \stackrel{kds_{i(j)}}{\rightarrow} N_d(j) + Q(r,j) i = 1, 2, \dots$
	$N(0,j) \stackrel{kds(j)}{ ightarrow} N_d(j)$
	$N_H(0,j) \stackrel{kds(j)}{\to} N_d(j)$
Reactions with poisons	$N_i(r,j) + I_m \stackrel{kdl(j)}{\rightarrow} N_{dlH}(0,j) + Q(r,j) i = 1, 2, \dots$
	$N_H(0,j) + I_m \stackrel{kdl(j)}{\rightarrow} N_{dlH}(0,j)$
	$N(0,j) + I_m \stackrel{kdI(j)}{ ightarrow} N_{dI}(0,j)$

- 1. Temperature gradients and radial concentrations in the reactor are negligible.
- 2. Elutriation of solids from the top of the reactor is neglected.
- 3. Overall movement direction of polymer particles is assumed to be downwards in both phases.
- 4. Constant mean particle size is considered through the bed.
- 5. The heat and mass transfer resistances between the emulsion gas and solid polymer particles are negligible, which is low to moderate catalyst activity.
- 6. Reaction occurs in both emulsion and bubble phases.
- 7. Catalyst is fed continuously into the bed as pre-polymer.

2.2.2. Modified dynamic model

In the present work, a modified dynamic two-phase model is developed. Solid entrainment at the top of the reactor has been taken into account since there are cases where elutriation rate cannot be ignored. Normally, most of the granular particles remain in the bed while the smaller ones will leave the reactor with the fluidizing gas. However, where velocities are several times greater than the terminal velocity, coarse particles can also be entrained from the bed [25]. This phenomenon is called particle carry over or particle entrainment, and is very important in the design and operation of FBRs. Elutriation takes place in the cyclone outside the FBR, and the solids are separated from the gas, reentering the reactor after some processing. This shows that in cases were particle entrainment occurs, it is vital to consider their effect on the process. As a result, in the present study, solid entrainment was considered in the model for mass and energy balances to make the results more realistic to the real data. An important property of any polymer grade is its MWD. It has been included in the model in order to check the validation of the model in comparison with industrial data.

In this model, the polymerization reaction is assumed to occur in both emulsion and bubble phases. Equations that are needed to calculate the heat and mass transfer coefficients, velocities in bubble and emulsion phase, and other useful parameters in the two-phase model are listed in Table 4. A concise list of assumptions used in the dynamic two-phase model is as follows: 243

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Table	3
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Reaction rate constants for polyethylene copolymerization [24].

	1 5 5	1 5		
Reaction	Rate constant	Unit	Site type 1	Site type 2
Formation	$k_f(j)$	s^{-1}	1	1
Initiation	$k_{i_1}(j)$	L/kmol s	1	1
	$k_{i_2}(j)$	L/kmol s	0.14	0.14
	$k_{h_1}(j)$	L/kmol s	1	1
	$k_{h_2}(j)$	L/kmol s	0.1	0.1
	$k_{h_r}(j)$	L/kmol s	20	20
Propagation	$k_{p_{11}}(j)$	L/kmol s	85	85
	$k_{p_{12}}(j)$	L/kmol s	2	15
	$k_{p_{21}}(j)$	L/kmol s	64	64
	$k_{p_{22}}(j)$	L/kmol s	1.5	6.2
Transfer	$k_{fm_{11}}(j)$	L/kmol s	0.0021	0.0021
	$k_{fm_{12}}(j)$	L/kmol s	0.006	0.11
	$k_{fm_{21}}(j)$	L/kmol s	0.0021	0.001
	$k_{fm_{22}}(j)$	L/kmol s	0.006	0.11
	$k_{fh_1}(j)$	L/kmol s	0.088	0.37
	$k_{fh_2}(j)$	L/kmol s	0.088	0.37
	$k_{fr_1}(j)$	L/kmol s	0.024	0.12
	$k_{fr_2}(j)$	L/kmol s	0.048	0.24
	$k_{fs_1}(j)$	L/kmol s	0.0001	0.0001
	$k_{fs_2}(j)$	L/kmol s	0.0001	0.0001
Deactivation	$k_{ds}(j)$	s^{-1}	0.0001	0.0001
	$k_{dI}(j)$	L/kmol s	2000	2000
Impurity	$k_a(j)$	s ⁻¹	0.0003	0.0003

- 1. The emulsion phase is considered to be completely mixed and not at the minimum fluidization condition.
- 2. Polymerization reactions are assumed to take place in both emulsion and bubble phases.
- 3. The bubbles are considered to be a sphere of constant dimensions and pass with unchanging velocity through the bed at plug flow condition.
- 4. Resistance of heat and mass transfer among gas and solid in emulsion and bubble phases are neglected.
- 5. Radial gradients for concentration and temperature in the reactor are neglected as a result of strict mixing brought about by the up-flowing gas.
- 6. Uniform particle size is considered all over the bed.
- 7. Solids entrainment is considered at the topmost part of the bed.

The mass balances obtained based on the assumptions of this model are as follows:

For emulsion phase:

$$M_{i]_{e,(in)}}U_{e}A_{e} - [M_{i}]_{e}U_{e}A_{e} - R_{\nu}\varepsilon_{e}[M_{i}]_{e} + K_{be}([M_{i}]_{b}$$

$$- [M_{i}]_{e})V_{e}\left(\frac{\delta}{1-\delta}\right) - (1-\varepsilon_{e})R_{i_{e}} - \frac{K_{e}V_{e}\varepsilon_{e}A_{e}[M_{i}]_{e}}{W_{e}}$$

$$= \frac{d}{dt}(V_{e}\varepsilon_{e}[M_{i}]_{e})$$
(3)

For bubble phase:

$$\begin{split} &[M_i]_{b,(in)} U_b A_b - [M_i]_b U_b A_b - R_{\nu} \varepsilon_b [M_i]_b - K_{be} ([M_i]_b - [M_i]_e) V_b \\ &- (1 - \varepsilon_b) \frac{A_b}{V_{PFR}} \int Ri_b dz - \frac{K_b V_b \varepsilon_b A_b [M_i]_b}{W_b} \\ &= \frac{d}{dt} \left(V_b \varepsilon_b [M_i]_b \right) \end{split}$$

$$(4)$$

The energy balances are expressed as follows: For emulsion phase:

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Please cite this article in press as: M.R. Abbasi et al., Dynamic modeling and Molecular Weight Distribution of ethylene copolymerization in an industrial gas-phase Fluidized-Bed Reactor, Advanced Powder Technology (2016), http://dx.doi.org/10.1016/j.apt.2016.05.014

Moments equations resulted from Table 1. Table 2

 $k_{h_{1}}(j)N_{H}(0,j)[\text{AIEt}_{3}] + \left\{ [M_{7}]k_{p_{77}}(j)2Y(1,j) - Y(0,j) \right\} + \left\{ Y(0,j) - Y(2,j) \right\} \left\{ k_{p_{17}}(j)[M_{7}] + k_{f_{7}}(j)[\text{AIEt}_{3}] \right\} - Y(2,j) \left\{ k_{p_{1}}(j)[H_{2}] + k_{d_{2}}(j) + k_{d_{3}}(j)[H_{1}] + k_{d_{3}}$

n = 0, 1, 2

 $(j)[\mathsf{AIEt}_3] + k_{fh_1}(j)[H_2] + k_{fs_1}(j) + k_{ds}(j) + k_{dl}(j)[I_m] \} - X(n,j)\frac{R_n}{V_n}$

 $= \{Y(n,j) - N_T(1,j)\}\{k_{fm_T}(j)[M_T] + k_{fr_T}$

Note: Rate constants with subscript *T* and

 $k_{h_T}(j)N_H(0,j)\} +$

 $\frac{N(1j)}{dt} = [M_T] \{ k_{i_T}(j) N(0,j) + i$ $\frac{Y(2J)}{dt} = [M_T] \{ k_{i_T}(j) N(0, j) + i$ π are pseudo-kinetic rate constants. The full

description and calculation method is given by McAuley et al. [24]

 $\{k_{h_1}(j)N_H(0,j)\} + \{k_{h_1}(j)N_H(0,j)](\text{AE}_{13}] + [M_T]\{k_{p_1}(j)Y(0,j) + \{Y(0,j) - Y(1,j)\}\{k_{h_{11}}(j)[M_T] + k_{f_1}(j)(\beta_{12}]_{13}]\} - Y(1,j)\{k_{h_{11}}(j)[H_{23}] + k_{f_2}(j) + k_{d_2}(j) + k_{d_3}(j)(m_1] + \frac{k_{p_1}}{k_{p_1}}\}$

 $\frac{dY(0,j)}{dt} = [MT] \{ k_{ft}(j)N(0,j) + k_{h_{f}}(j)N_{H}(0,j) \} + k_{h_{f}}(j)N_{H}(0,j) [AIEt_{3}] - Y(0,j) \{ k_{h_{f}}(j)[H_{2}] + k_{h_{f}}(j) + k_{dt}(j) + k_{dt}(j)[I_{m}] + \frac{R_{f}}{P_{p}} \} \}$

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Table 4

Hydrodynamic equations used in the model.

Parameter	Formula	Reference
Minimum fluidization velocity	$\operatorname{Re}_{mf} = \left[29.5^2 + 0.375 Ar\right]^{0.5} - 29.5$	[26]
Bubble velocity	$U_b = U_0 - U_{mf} + U_{br}$	[27]
Bubble rise velocity	$U_{br} = 0.711 (gd_b)^{0.5}$	[27]
Emulsion velocity	$U_e = \frac{U_0 - U_b}{(1 - \delta)}$	[28]
Bubble diameter	$d_b = d_{b0} [1 + 27(U_0 - U_e)]^{0.33} (1 + 6.84H) d_{b0} = 0.0085$ (for GeldartB)	[29]
Mass transfer coefficient	$k_{be} = \left(\frac{1}{k_{be}} + \frac{1}{k_{ce}}\right)^{-1}$	[27]
	$K_{bc} = 4.5 \left(\frac{U_c}{d_b} ight) + 5.85 \left(\frac{D_b^{S,5} g^{23}}{d_b^{1/25}} ight)$	
	$K_{ce} = 6.77 \left(rac{D_g \hat{v}_e U_{br}}{d_b} ight)$	
Heat transfer coefficient	$H_{be} = \left(\frac{1}{H_{bc}} + \frac{1}{H_{cc}}\right)^{-1}$	[27]
	$H_{bc} = 4.5 \left(\frac{U_e \rho_g C_{pg}}{d_b} \right) + 5.85 \frac{(k_g \rho_g C_{pg})^{0.5} g^{0.25}}{d^{1.25}}$	
	$H_{ce} = 6.77 \left(\rho_g C_{pg} k_g \right)^{0.5} \left(\frac{c_e U_{br}}{3} \right)^{0.5}$	
Bubble phase fraction emulsion	$\delta = 0.534 \left[1 - \exp\left(\frac{U_0 - U_{mf}}{0.413}\right) \right]$	[30]
Emulsion phase porosity	$\varepsilon_e = \varepsilon_{mf} + 0.2 - 0.059 \exp\left(-\frac{U_0 - U_{mf}}{0.429}\right)$	[30]
Bubble phase porosity	$\varepsilon_b = 1 - 0.146 \exp\left(-\frac{U_0 - U_{mf}}{4.439}\right)$	[30]
Volume of polymer phase in the emulsion phase	$V_{Pe} = AH(1 - \varepsilon_e)(1 - \delta)$	[22]
Volume of polymer phase in the bubble phase	$V_{Pb} = AH(1 - \varepsilon_b)\delta$	[22]
Volume of the emulsion phase	$V_e = A(1-\delta)H$	[22]
Volume of the bubble phase	$V_b = A\delta H$	[22]
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Table 5

Operating conditions for petrochemical complex 1.

Parameter	BP LL0209	BP HD3840	BP HD5218	BP HD6070
$D_t(\mathbf{m})$	4.8	4.8	4.8	4.8
<i>H</i> (m)	14.5	14.5	14.5	14.5
$d_p(\mu m)$	1145	1049	1061	965
T (°C)	317	310	313	316
P (bar)	20	19.91	19.85	19.99
U_0 (m/s)	0.57	0.56	0.54	0.55
Ethylene concentration (%)	40	40	40	34
1-Butene concentration (%)	17	6.43	2.36	0.34
Hydrogen concentration (%)	9	16	30	23.46
Nitrogen concentration (%)	34	37.57	27.64	42.2
Catalyst feed rate (g/s)	0.2	0.2	0.2	0.2

$$\begin{split} &U_e A_e (T_{e \cdot (in)} - T_{ref}) \sum_{i=1}^m [M_i]_{e,(in)} C_{pi} - U_e A_e (T_e - T_{ref}) \sum_{i=1}^m [M_i]_e C_{pi} \\ &- R_\nu (T_e - T_{ref}) \left(\sum_{i=1}^m \varepsilon_e C_{pi} [M_i]_e + (1 - \varepsilon_b) \rho_{pol} C_{p,pol} \right) \\ &+ (1 - \varepsilon_e) R_{pe} \Delta H_R - H_{be} V_e \left(\frac{\delta}{1 - \delta} \right) (T_e - T_b) \\ &- V_e \varepsilon_e (T_e - T_{ref}) \sum_{i=1}^m C_{pi} \frac{d}{dt} ([M_i]_e) \\ &- \frac{K_e A_e}{W_e} (T_e - T_{ref}) \left(\sum_{i=1}^m \varepsilon_e C_{pi} [M_i]_e + (1 - \varepsilon_b) \rho_{pol} C_{p,pol} \right) \\ &= \left(V_e \left(\varepsilon_e \sum_{i=1}^m C_{pi} [M_i]_e + (1 - \varepsilon_e) \rho_{pol} C_{p,pol} \right) \right) \frac{d}{dt} (T_e - T_{ref}) \end{split}$$

For bubble phase:

$$\begin{split} &U_{b}A_{b}(T_{b\cdot(in)} - T_{ref})\sum_{i=1}^{m}[M_{i}]_{b,(in)}C_{pi} - U_{b}A_{b}(T_{b} - T_{ref})\sum_{i=1}^{m}[M_{i}]_{b}C_{pi} \\ &- R_{\nu}(T_{b} - T_{ref})\left(\sum_{i=1}^{m}\varepsilon_{b}C_{pi}[M_{i}]_{b} + (1 - \varepsilon_{e})\rho_{pol}C_{p,pol}\right) \\ &+ (1 - \varepsilon_{b})\frac{A_{b}H_{R}}{V_{PFR}}\int R_{pb}dz + H_{be}V_{b}(T_{e} - T_{b}) \\ &- V_{b}\varepsilon_{b}(T_{b} - T_{ref})\sum_{i=1}^{m}C_{pi}\frac{d}{dt}([M_{i}]_{b}) \\ &- \frac{K_{b}A_{b}}{W_{b}}(T_{b} - T_{ref})\left(\sum_{i=1}^{m}\varepsilon_{b}C_{pi}[M_{i}]_{b} + (1 - \varepsilon_{b})\rho_{pol}C_{p,pol}\right) \\ &= \left(V_{b}\left(\varepsilon_{b}\sum_{i=1}^{m}C_{pi}[M_{i}]_{b} + (1 - \varepsilon_{b})\rho_{pol}C_{p,pol}\right)\right)\frac{d}{dt}(T_{b} - T_{ref}) \end{split}$$

Solid elutriation constants are obtained from Rhodes [25] and are as follows:

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(5)



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Fig. 2. Molecular Weight Distribution of the produced LLDPE.





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$$K_{e} = 23.7 \rho_{g} U_{0} \frac{A}{W_{e}} \exp\left(\frac{-5.4U_{t}}{U_{0}}\right)$$

$$(7) \qquad U_{t}^{*} = \left[18(d_{p}^{*})^{-2} + (2.335 - 1.744 \varnothing_{s})(d_{p}^{*})^{-0.5}\right]^{-1}$$

$$(11)$$
For $0.5 < \varnothing_{s} \le 1$,

(9)

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$$K_b = 23.7 \rho_g U_0 \frac{A}{W_b} \exp\left(\frac{-5.4U_t}{U_0}\right) \qquad (8)$$

$$d_p^* = d_p \left(\frac{1}{W_b}\right) = \frac{1}{W_b} \left(\frac{1}{W_$$

$$W_e = AH(1 - \varepsilon_e)\rho_{pol}$$

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$$W_b = AH(1 - \varepsilon_b)\rho_{pol}$$
(10)

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$$U_t = U_t^* \left[\mu \rho_g^{-2} (\rho_{pol} - \rho_g) g \right]^{0.33}$$
(11)

$$U_t^* = \left[18(d_p^*)^{-2} + (2.335 - 1.744 \varnothing_s)(d_p^*)^{-0.5}\right]^{-1}$$
(12) 300

For
$$0.5 < \emptyset_s \le 1$$
,
 $d_p^* = d_p \Big[\mu^{-2} \rho_g (\rho_{pol} - \rho_g) g \Big]^{0.33}$
(13) 304

These equations can be solved using the following initial conditions:

$$[M_i]_{b,t=0} = [M_i]_{in} \tag{14} \tag{14}$$

$$T_{b,t=0} = T_{in}$$
 (15) 312

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Fig. 4. Polydispersity Index comparison of four grades of polyethylene with industrial and literature data.

 $[M_i]_{e,t=0} = [M_i]_{in}$ (16)315

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$$T_{e,t=0} = T_{in}$$
 (17)

319 These sets of equations have been coded and solved in less than 3 s using MATLAB and Simulink. 320

3. Results and discussion 321

322 In order to show how the model responds when tested with real 323 data and to validate it, the operating conditions listed in Table 5 324 were used in performing the simulation study. The data of four different grades of polyethylene produced at a petrochemical com-325 pany reported by Kiashemshaki et al. [19] is used to both 326 327 validate and compare the model. Unless mentioned otherwise,

the results are based on the operating conditions for grade BP LL0209 as listed in Table 5.

The conditions mentioned in this table are common to produce these grades of polyethylene in industrial reactors. Polymer properties such as molecular weight, PDI and Melt Flow Index (MFI), which are crucial to estimate the quality of a given polymer, have been calculated based on the kinetic model used in this work. Using methods described by McAuley et al. [24], the weight average and number average molecular weight of polymer can also be calculated.

Polymers are made of many repeated units (monomers) which are chemically attached and make very long chains. Having a perception of polymer chain length is obligatory to comprehend the physical properties of a polymer. Chain length is frequently denoted as the molecular weight of the polymer chain, which is correlated to the number of monomers connected in the chain and the relative molecular mass of the monomers. Nevertheless. all artificial polymers are polydisperse, which means that the length of polymer chains are unequal, and as a result, instead of being a single value, the polymer has a distribution of molecular weights and chain lengths. Consequently, some average molecular weight must be calculated from the molecular weights of all the chains in the polymer sample to define the molecular weight.

Fig. 2 demonstrates how the simulation result, based on the presented model and industrial data of a LLDPE, predicts a narrow MWD for the polymer.

The number average molecular weight is defined as the arithmetic mean of all the molecular weights of the polymer chains in the sample, given by:

$$\overline{M}_{n} = \frac{\sum N_{i}M_{i}}{\sum N_{i}}$$
(18)

where N_i is the number of chains of that molecular weight and M_i is 360 the molecular weight of a chain. \overline{M}_n is measured by approaches that 361 define the number of molecules in a sample of a particular weight 362 and can be predicted by means of polymerization mechanisms. If 363 \overline{M}_n is mentioned for a certain MWD, it means that identical num-364 bers of molecules are present on both sides of \overline{M}_n in the distribu-365 tion. On the other hand, the weight average molecular weight is 366 367 given by:





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Fig. 6. Evolution of polymer production rate over time in the FBR.





$$\overline{M}_{w} = \frac{\sum N_{i} M_{i}^{2}}{\sum N_{i} M_{i}}$$
(19)

Compared to \overline{M}_n , \overline{M}_w takes the molecular weight of a chain into 371 consideration to decide how it contributes to the average molecu-372 373 lar weight. The larger the chain gets, the effect of chain on M_w 374 increases. Instead of the number of molecules, weight average 375 molecular weight is defined by procedures which measure the 376 molecular size, such as through light scattering techniques. If \overline{M}_w 377 is mentioned for a certain MWD, it means that identical weight 378 of molecules is present on both sides of M_w in the distribution. These values are illustrated in Fig. 3. This figure shows that it takes 379 less than an hour for the number average and weight average 380 381 molecular weights of polymer to reach a constant value, since the molecular chain length grows rapidly during this time. As382shown in this figure, the ultimate amount of weight average383molecular weight reaches almost 99,000 kg/mole.384

The PDI of a polymer is expressed as the weight average molecular weight to number average molecular weight proportion, and is used as a parameter to tell how broad a polymer MWD and is given by:

$$PDI = \frac{\overline{M}_w}{\overline{M}_n} \tag{20}$$

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If a polymer has bigger PDI value, the polymer molecular weight is broader. A polymer with PDI = 1 in which all the chain lengths are equivalent (such as a protein) is called a monodisperse poly-394

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Fig. 8. Emulsion phase temperature comparison with industrial and literature data for the four different grades of polyethylene after reaching steady state.

mer. The narrowest artificial polymers built so far which are used
for calibration have PDI of 1.02–1.10. While chain reactions produce PDI values between 1.5 and 20 while step polymerization
reactions usually PDI values of around 2.0 [26]. The PDI profile
throughout the polymerization process is similar to the average
molecular weight.

Another important property of a polymer is MFI. It is an analysis method that controls how easily a plastic material flows and is a very important test for quality assurance. In order to measure MFI, the amount of a polymer that flows from a standard instrument over a timed interval is weighed.

The relationship between the molecular weight of polyethylene and its MFI is based on the type proposed by McAuley et al. [24], whose constants have been modified to fit the actual data and is given by the following equation:

$$MFI = 3.346 \times 10^{17} \overline{M}_{W}^{-3.472}$$
(21) 412

The steady state value of PDI and MFI under the operating conditions given in Table 5 are 4.14 and 0.98 g/10 min respectively. Fig. 3 also shows the evolution of PDI and MFI with time in the reactor.

For model verification purposes, the results of the model presented in this work for PDI and MWD have been compared with the actual plant data and results from the study done by Kiashemshaki et al. [19]. The parity plot of Fig. 4 shows that the current dynamic model has been able to predict the PDI of LLDPE very accurately and very close to the work of Kiashemshaki et al. [19]. As mentioned by the authors, the difference in the calculated PDIs of HDPEs could be due to considering the same catalyst properties for all the grades. However, catalysts are produced in different batches in this plant and could have slightly different properties and rate constants and hence can result in the deviation from the actual PDIs.

Furthermore, the calculated steady state MWD has been com-429 pared with the literature [19] and actual MWD data points for a 430 LLDPE (BP LL0209) and a HDPE (BP HD5218) grade in Fig. 5 for 431 comparison. The actual data have been produced using the GPC 432 data provided by the petrochemical complex. These two figures 433 are produced by calculating and plotting the derivative of cumula-434 tive weight fraction against the logarithm of weight average 435 molecular weight, which is a typical GPC output. As can be seen, 436 aside from the slight differences, there is a very good agreement 437 between all sets of data for both cases of LLDPE and HDPE grades. 438

Although taking solids elutriation into account in the present 439 model leads to predicting the polymer properties such as MFI, 440 PDI and MWD accurately, its main advantages lie in calculating 441 the process parameters such as production rate and reactor tem-442 perature more precisely. This is due to the improved dynamic mass 443 and energy balance equations which consider solids entrainment 444 and essentially improves the model to comply more with the real 445 world process data. In addition, this model is able to predict the 446 dynamic behavior of the fluidized bed reactor and can also be used 447 for control study and designing an efficient control system for this 448 highly nonlinear process. 449

Polymer production rate during residence time in the FBR is given by Eq. (2) and is shown in Fig. 6. This figure shows the evo-





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Table 6

Operating conditions for petrochemical complex 2.

Parameter	BP LLO2	209												
Time	8:00	8:30	9:00	9:30	10:00	10:30	11:00	11:30	12:00	12:30	13:00	13:30	14:00	14:30
<i>H</i> (m)	19.76	19.77	19.74	19.79	19.87	19.87	19.83	19.6	19.91	20.12	20.13	20.04	20.11	20.12
d_p (µm)	986	986	986	986	986	986	986	986	986	986	986	986	986	986
P (bar)	21.8	21.81	21.81	21.8	21.83	21.79	21.81	21.81	21.74	21.74	21.73	21.77	21.75	21.76
<i>U</i> ₀ (m/s)	0.634	0.635	0.636	0.635	0.635	0.633	0.634	0.633	0.634	0.634	0.631	0.629	0.631	0.632
Ethylene (%)	37.07	36.9	36.88	36.95	37.02	36.96	36.8	36.67	36.63	36.6	36.44	36.54	36.45	36.46
1-Butene (%)	13.73	13.75	13.73	13.71	13.69	13.67	13.63	13.64	13.55	13.55	13.57	13.54	13.56	13.57
Hydrogen (%)	4.57	4.58	4.62	4.62	4.6	4.6	4.65	4.65	4.63	4.65	4.69	4.66	4.69	4.68
Inert gas (%)	36.39	36.4	36.32	36.38	36.47	36.32	36.41	36.38	36.42	36.49	36.7	36.57	36.7	36.7
Catalyst feed rate (g/s)	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661	0.661



Fig. 10. Evolution of mean ethylene and 1-butene concentration throughout the bed during residence time in the FBR.

452 lution of production rate from the start-up moment when Ziegler-Natta catalyst enters the reactor, and reaction starts until the time 453 that solid particles settle in the FBR, and the fluidized bed moves to 454 the steady state condition and the production rate becomes steady 455 (Fig. 1). This figure also shows polymer production rate in both the 456 bubble and emulsion phases during the polymer residence time in 457 the reactor. The calculated overall production rate soars from 458 459 almost 7 t/h in the first hour to almost 10 t/h in the second hour. and becomes steady at 13.44 t/h after nearly 5 h of production. 460 461 To show the model accuracy and validate the results, the produc-462 tion rate has been plotted against both the industrial data and 463 the model of Kiashemshaki et al. [19]. The horizontal line showing actual data is the amount for the steady state production rate in 464 the industrial FBR. As can be seen, the model was able to accurately 465 466 predict the steady state production rate and stabilize very close to 467 this data with a deviation of 0.4 tones. The figure also illustrates 468 that nearly 60% of the polymer is produced within the emulsion 469 phase and almost 40% of the total polymer production takes place 470 within the bubble phase. The 20% increase in the production rate in 471 bubble phase in comparison with Kiashemshaki et al. [19] is due to 472 the introduction of recycled elutriated solids into the reactor. This 473 predictably increases the amount of catalysts in the bubble phase 474 and leads to higher production in this phase. Since most of the cat-475 alysts are within the emulsion phase, less reaction rate or polymer 476 production in the bubble phase is inevitable.

The evolution of emulsion phase temperature during the poly-477 mer residence time in the reactor is shown in Fig. 7 for the four 478 different grades of polymer. All grades start from temperatures 479 given in Table 5 and continue to increase with different slopes, 480 and becomes steady after almost 5 h. For example, it is estimated 481 that the temperature of the LLDPE reaches 78 °C after getting 482 steady around the fifth hour and remains at that temperature. 483 The emulsion phase temperature grows rapidly in less than an 484 hour and reaches 58 °C after one hour from a temperature of 485 44 °C in the beginning, since the polymerization reaction is 486 exothermic and this graph illustrates this clearly. The final steady 487 state temperatures of this grades are compared with both indus-488 trial data and the work of Kiashemshaki et al. [19] in the parity plot 489 of Fig. 8. The lower LLDPE temperature compared with the HDPE 490 temperatures and literature data is due to the higher superficial 491 gas velocity of LLDPE grade compared to HDPE grade. There is a 492 direct relation between superficial gas velocity and the monomer 493 residence time in the reactor, heat removal rate from the reactor, 494 particle mixing and fluidization conditions. In fact, by increasing 495 the superficial gas velocity, gas passes faster through the bed. As 496 a result, more solid particles carry over, the amount of catalysts 497 and polymer particles available in the reactor bed will be reduced, 498 and some monomers may bypass the catalysts, and therefore, reac-499 tion extent will be reduced and lead to a reduction in the reaction 500 rate. Less reactions in this exothermic reaction means lower tem-501

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Fig. 11. Relation between 1-butene concentration of the feed and polymer density.

peratures. Since particle entrainment is considered in this model, 502 503 this will justify the lower temperature results of this model in 504 LLDPE grade compared to those obtained by Kiashemshaki et al. 505 [19]

506 Another advantage of the current model is calculating temper-507 atures dynamically. Unlike the presented model in this paper, Kiashemshaki et al.'s [19] work is not dynamic in terms of calculating 508 509 both temperature and comonomer concentration. A dynamic 510 model can have the advantage that it can be used as a basis in pro-511 cess control studies to test different approaches to control polymer 512 properties and reactor parameters based on parameters like inlet 513 gas compositions, catalyst input rate, gas superficial velocity, and 514 reactor pressure. Moreover, solids elutriation is another phe-515 nomenon in FBRs which cannot be neglected, and is included in this work to make the model be more realistic. Fig. 8 shows that 516 517 the temperatures calculated in this work are more accurate and 518 closer to actual data.

519 Table 6 shows the operating conditions of another industrial 520 polyethylene production reactor in a second petrochemical complex during one working shift. The operational data and resulting 521 temperature data were collected using plant distributed control 522 system. To further validate the model, it was tested using this data-523 524 set. Calculated reactor temperatures have been compared with real 525 data in Fig. 9. The model has again been capable of accurately pre-526 dicting reactor temperature for this grade of polymer. The average 527 error for this dataset is 0.6 percent deviation from the industrial 528 data, which is small in the engineering context. Nevertheless, con-529 sidering resistance of heat transfer among gas and solids in both phases, radial temperature gradients in the reactor and particle 530 size distributions could further improve the model, which leads 531 to better prediction of reactor parameters and polymer properties. 532 533 However, this will increase model complexity and computational 534 efforts.

The evolution of mean monomer concentrations throughout the 535 bed during the residence time in the reactor are shown in Fig. 10. It 536 537 is clear that the time of getting into a steady state is the same as 538 production rate and temperature profile graphs. As can be seen, 539 ethylene as monomer and 1-butene as comonomer are consumed 540 through the copolymerization reaction in order to produce the 541 polymer. As a result, their concentrations decrease exponentially

during the first 5 h of the residence time before going flat in the steady zone.

Another imperative property of the polymer is its density. Sev-544 eral polymer grades for different applications need to have specific 545 densities. Since this model is dynamic, it is capable to be used in 546 future polymer density control studies. However, it is very compli-547 cated to find the correlation between density and polymer struc-548 ture. Density could be altered by both the length and number of 549 the short chain branches and to a small extent by the polymer 550 molecular weight [27]. McAuley et al. [24] developed an experi-551 mental equation to relate the amount of comonomer in linear poly-552 ethylene to its density:

$$\rho = 0.966 - \alpha C_x^\beta \tag{22}$$

where α and β are parameters which depend on comonomer, and C_{x} 557 is the comonomer mole percent in the polymer. α and β have been 558 fitted at 0.02386 and 0.514 for butane grade polymers. Fig. 11 559 shows the correlation between 1-butene concentration of the feed 560 and the density calculated from Eq. (22). Naturally, increasing 1-561 butene concentration leads to lower density values and it can be 562 used as a manipulated variable in future process control studies 563 to regulate the polymer density value. 564

4. Conclusion

A comprehensive two-phase model was developed to predict industrial scale gas-phase ethylene copolymerization reactors. The model considers solid entrainment in the FBR modeling. The hydrodynamic model coupled with a kinetic copolymerization model (ethylene and 1-butene) provides a more detailed understanding of the system.

The model was capable of predicting vital reactor parameters 572 like rate of polymer production and polymer temperature. More-573 over, the kinetic model was capable of predicting polymer proper-574 ties such as PDI, average molecular weights and MWD of the 575 polymer. The PDI, MWD, production rate and reactor temperature 576 results of this model were compared with actual plant data and lit-577 erature to show the data agreement. This model provides a tool to 578 study the operational, hydrodynamics and kinetic parameters on 579 reactor performance in addition to polymer properties, and can

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be used as a base for control studies to regulate properties like PDI,MFI or density of a polymer in future works.

583 Acknowledgements

The authors would like to thank the support of the Research Council of University of Malaya under High Impact Research Grant (UM.C/625/1/HIR/MOHE/ENG/25) and Postgraduate Research Grant (PPP) project No. PG131-2014A.

588 Appendix A. Supplementary data

589 Supplementary data associated with this article can be found, in 590 the online version, at http://dx.doi.org/10.1016/j.apt.2016.05.014.

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