Synthesis And Characterization Of LiNi_{1-x}Zn_xVO₄

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Abstract

LiNi_{1-x}Zn_xVO₄ was synthesized at x = 0.0, 0.2, 0.4 and 0.8. by a polymer hybrid sol gel method. In this work acetates of lithium (C₂H₃O₂Li), nickel (C₂H₃O₂Ni) and zinc (C₂H₃O₂Zn) materials were dissolved in distilled water, stirred and heated continuously until a clear green solution was obtained. At this stage, a solution of chitosan in 1 % acetic acid was added. The was sintered at 700 °C for 3 hours. The crystallite size, lattice constant and surface peaks without impurities. Zinc substituted LiNiVO₄. X-ray diffraction (XRD) exhibits 0.0 to x = 0.4. Crystallite size increases again from x = 0.4 to x = 0.8. Cyclic voltammetric (CV) studies were performed.

Keywords: substitution, cathode materials, sol-gel method

1. INTRODUCTION

Lithium ion batteries are given much attention due to its excellent properties such as high voltage, high energy density and good cycling charge characteristics [1]. LiNiVO₄ is known as potential cathode material for its use in the lithium ion batteries. It exhibits 4.8 V versus Li⁺/Li in lithium ion cell. LiNiVO₄ is the first cathode material with an inverse spinel structure that has been used in Li-ion batteries [2]. The presence of nickel atoms in the spinel structure has found to be an important factor in the voltage behaviour of this material [3]. Substitution is one of the methods to improve the capacity of the batteries [4-6].

Doped cathode materials have shown improvements on the electrochemical performance of the battery. Examples include doped LiNiO₂ [5], LiCoO₂ [7], LiMn₂O₄ [8-11] Structural and vibrational studies on Co substituted LiNiVO₄ has been carried out by Julien et al. [12]. This work focused mainly on the structural analysis. In the present work, zinc substituted LiNiVO₄ was prepared by chitosan modified sol-gel method. The structure analyses were carried using XRD and TEM. Cyclic voltammetric studies were performed on zinc substituted LiNiVO₄.

2. EXPERIMENTAL

2.1 The Preparation and Characterization of $LiNi_{1-x}Zn_xVO_4$

Chitosan modified sol-gel method was used to synthesize $LiNi_{1-x}Zn_xVO_4$ at x = 0.0, 0.2,0.4 and 0.8. Lithium acetate (C2H3O2Li), nickel acetate (C2H3O2Ni) and zinc acetate (C₂H₃O₂Zn) with ammonium metavanadate (NH₄VO₃) were used as starting materials to prepare LiNi1-xZnxVO4. The given starting materials were weighed for different compound ratios as x varies in LiNi1-_xZn_xVO₄. These materials were dissolved in distilled water, stirred and heated continuously until a clear green solution was obtained. At this stage, a solution of

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chitosan in 1 % acetic acid was added. The solution was continuously stirred and heated until it dried. The obtained solid was sintered at 700 °C for 3 hours to eliminate impurities.

The diffraction intensity at different glancing angles for LiNiVO4 and zinc substituted LiNiVO4 were measured by using X-ray diffractometer (Bruker AXS) with operating voltage and current of 40 kV and 40 mA respectively. The X-ray wavelength is 1.5406 Å and the glancing angles were in the range $5^{\circ} \le 2\theta \le 80^{\circ}$ with a step size of 0.04°. The samples were characterized using Leo Libra 120 transmission electron microscope (TEM) instrument. Electrochemical tests were performed at Autolab. Cathode was prepared by 20 mg active material with 8 mg TAB (Teflonized Acetyle Black). Lithium metal was used as anode. The electrolyte used was 1M LiPF₆ EC/DEC (1:2).

3. RESULTS AND DISCUSSION

Fig. 1 shows the XRD patterns for zinc substituted LiNiVO4 samples. All the substituted samples were sintered at 700 °C to eliminate the impurities. The peaks are similar to LiNiVO4. All the peaks correspond to Fd3m space group with face centered cubic structure. The substitution with zinc did not change the structure of the material. There are no peaks that correspond to impurities such as NiO (at $2\theta = 43.5^{\circ}$) and Li_3VO_4 (at $2\theta = 16.2^\circ$, 21.5° , 22.8° and 24.3°) which was found in some works on LiNiVO₄ [13-14]. This confirms that sintering at 700 °C enables elimination of impurities that could defect the structure of the product and result in poor performance of LiNiVO4.

Crystallite size (d) of the samples was calculated using Scherrer's equation [15-16]:

$$d = \frac{0.9\lambda}{\beta\cos\theta} \tag{1}$$

where β is FWHM at the most intense diffraction peak (311) plane in radians, θ is wavelength of CuK α X- radiation. The results are presented in the Table 1.



Fig. 3 X-ray diffraction of LiNi_{1-x}Zn_xVO₄ sintered at 700 °C

Table 1	Crystallite	size	of	the	Zn	substituted
LiNiVO4						

	Crystallite size (nm)		
7.10			
x in $LiNi_{1-x}Zn_xVO_4$	97.2		
0.0	86.1		
0.2	49.1		
0.4	65.8		
0.8	0010		

Table 1 indicates that the changes of crystallite size with the zinc substitution in LiNiVO₄. The crystallite size of the zinc substituted LiNiVO4 decreased markedly from x = 0.0 to x = 0.4, and it is increasing from x = 0.4 and x = 0.8. The values of calculated crystallite size for every samples proved that the particles are in the size of nanometers. The crystallite size of cathode materials is another factor that plays an important role in the electrochemical properties. Smaller crystallites with larger surface area are able to reduce diffusion length of Li⁺ ion. This results in better lithium ion intercalation which leads to improved electrochemical performance [1718]. TEM images as shown in Fig. 2 agree well with the calculated crystallite size. All the samples exhibit nanosized particles.





Fig. 2 TEM micrographs of (a) LiNiVO₄, (b) LiNi_{0.8}Zn_{0.2}VO₄, (c) LiNi_{0.6}Zn_{0.4}VO₄ and (d) LiNi_{0.2}Zn_{0.8}VO₄





The sizes of particles are in the range of 50 nm to 90 nm for LiNiVO₄. The particles of LiNi_{0.8}Zn_{0.2}VO₄ and LiNi_{0.6}Zn_{0.4}VO₄ are in the range of 40 nm to 70 nm and 30 nm to 50 nm respectively. The particles of LiNi_{0.2}Zn_{0.8}VO₄ are also nearly in the same range which is 30 nm to 60 nm.

Lattice constant showed a slight increase with substitution of zinc due to the different ionic radius of zinc and nickel [19]. When Zn^{2+} ion with larger ionic radius (0.88 Å) replaces Ni²⁺ ion with smaller ionic radius (0.83 Å), the lattice constant increased as tabulated in Table 2. Decrease of lattice constant at LiNi_{0.8}Zn_{0.2}VO₄ may be due to a different structure formed at higher zinc substitution [20].

Table 2 Lattice constant of zinc substituted LiNiVO4

x in LiNi _{1-x} Zn _x VO ₄	lattice constant (Å)
0.0	8.1964
0.2	8.1968
0.4	8.2140
0.8	8.2095

Cyclic voltammograms of the samples at scan rate 0.1 mV/s in the potential region of 3 to 4.8 V is shown in Fig 3.

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Fig. 3 Cyclic voltammograms of (a) LiNiVO4, (b) $LiNi_{0.8}Zn_{0.2}VO_4$, (c) $LiNi_{0.6}Zn_{0.4}VO$ and (d) LiNi0.2Zn0.8VO

Table 3 gives the values of oxidation and reduction peaks for substituted samples.

Table 3 Oxidation	and	reduction	peaks	of	zinc
substituted LiNiVO4			100		

x in	Oxidation	Reduction neaks (V)		
LiNi _{1-x} Zn _x VO ₄	peaks (V)	1.05		
0.0	4.71	4.05		
0.2	4.59	4.14		
0.2	4 56	4.15		
0.4	1.50	4 17		
0.8	4.54	4.17		

The oxidation voltage decreases with increase in zinc substitution. The reduction peak increases with substitution of zinc. This shows improved cycling stability and reversibility [21].

4. CONCLUSIONS

 $LiNi_{1-x}Zn_xVO_4$ was synthesized at x = 0.0, 0.2, 0.4 and 0.8 by chitosan modified sol gel method. The crystallite sizes of zinc substituted samples are smaller compared to LiNiVO₄. Cyclic voltammogram shows better cycling stability and reversibility.

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REFERENCES

- D. Zhang, B.S. Haran, A. Durairajan, R.E. White, Y. Podrazhansky and B.N. Popov. [1] J. Power Sources 91 (2000) 122
- G.T.K. Fey and K.S. Chen. J. Power [2]
- Sources 81-82 (1999) 467 G.T.K. Fey, K.S. Wang and S.M. Yang. J. [3]
- Power Sources 68 (1997) 159 G.T.K Fey, P. Muralidharan, C.Z. Lu,
- Y.D. Cho. Solid State Ionics 177 (2006) [4]
- M.Y. Song, C. Park, S. Yoon, H. Park and D.R. Mumm. Ceramics International 35 [5]
- Y. Ding, P. Zhang, Y. Jiang, J. Yin, Q. Lu (2009) 1145 and D. Gao. Materials Research Bulletin [6] 43 (2008) 2005

NATIONAL WORKSHOP ON FUNCTIONAL MATERIALS 2009

- [7] R. Sathiyamoorthy, R. Chandrasekaran,
 A. Gopalan and T. Vasudevan. Materials Research Bulletin 43 (2008) 1401
- [8] I.S. Jeong, J.U. Kim and H.B. Gu. J. Power Sources 102 (2001) 55
- [9] Z. Bakenov and I. Taniguchi. Solid State Ionics 176 (2005) 1027
- [10] Y.K. Yoon, C.W. Park, H.Y. Ahn, D.H. Kim, Y.S. Lee and J. Kim. J. Physics and Chemistry of Solids 68 (2007) 780

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- [11] K. Suryakala, K.R. Marikkannu, G. Paruthimal Kalaignan and T. Vasudevan. Ionics 13 (2007) 41
- [12] C. Julien, M. Massot and C. Perez-Vicente. Materials Science and Engineering B75 (2000) 6
- [13] G.T.K. Fey, J.R. Dahn, M. J. Zhang and W. Li. J. Power Sources 68 (1997) 549

- [14] G.T.K. Fey and W.B. Perng. Materials Chemistry and Physics 47 (1997) 279
- [15] T. Yi, C. Dai, K. Gao and X. Hu. J. Alloys and Compounds 425 (2006) 343
- [16] J. Molenda, W. Ojczyk and J. Marzec. J. Power Sources 174 (2007) 689
- [17] R. Dziembaj and M. Molenda. J. Power Sources 119-121 (2003) 121
- [18] M. Bahgat, F.E. Farghaly, S.M. Abdel Basir and O.A. Fowad. J. Materials Processing Technology 183 (2007) 117
- [19] C.H. Lu and S.W. Lin. J. Power Sources 97-98 (2001) 458
- [20] Y.T. Feng, W. Dianlong, G. Kun and H. Xinguo. Rare metals 26 (2007) 330
- [21] B.L. He, W.J. Zhou, Y.Y. Liang, S.J. Bao and H.L. Li. J. Colloid and Interface Science 300 (2006) 633

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[22] I. Kwon and M.Y. Song. Solid State Ionics 158 (2003) 103